Core−Shell−Corona-Structured Polyelectrolyte Brushes-Grafting Magnetic Nanoparticles for Water Harvesting

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S Supporting Information

[AB](#page-6-0)STRACT: [A novel sup](#page-6-0)erhydrophilic material, charged polymer brushes-grafted magnetic core−shell−corona composite nanoparticles (Fe₃O₄@SiO₂@PSPMA), was developed to harvest water through the hydration effect. Because of both the strong hydration capability and the good swelling performance, the negatively charged polymer brushes, PSPMA brushes, endow the composite nanoparticles with superhydrophilicity and a good water-absorbing performance like a sponge, while the magnetic Fe₃O₄ cores allow easy separation of Fe₃O₄@ $SiO₂(\omega$ PSPMA nanoparticles with absorbed water from oil/ water mixture under an external magnetic field. The functional particles have the capability of harvesting water droplets whether floating on an oil surface or in the oil. This water-

absorbing material uses selective wettability to harvest water and achieve oil−water separation and may be useful in finding novel approaches for recycling water from sewage and removing water in the petroleum industry.

KEYWORDS: charged polymer brushes, Fe₃O₄@SiO₂@PSPMA, magnetic nanoparticles, hydration effect, water harvesting, oil−water separation

1. INTRODUCTION

With increasing environmental pollution and water shortage, novel strategies to harvest and separate water from oil−water mixture are highly desired. Since water harvesting and oil/water separation is an interfacial problem, it is an effective and a simple strategy to design novel materials with special wettability.^{1−3} For oil/water separation, "oil-absorbing" types of materials with both superhydrophobic and superoleophilic properties [hav](#page-6-0)e been widely used to filtrate or absorb oils from water effectively.^{4−6} However, oil-absorbing materials are easily fouled by oils, and oils adhered are difficult to remove, leading to the marked [decr](#page-6-0)ease of separation efficiency as well as the secondary pollution. 7 In practical applications, it may be an alternative strategy to design "water-absorbing" materials for harvesting water [an](#page-6-0)d achieving oil/water separation. In comparison with traditional oil-absorbing materials, this novel water-absorbing material has completely opposite wettability and thus overcomes the easy-fouling limitation.^{8−10} Currently, oil−water separation methods mostly focus on the design of hierarchical structure.^{11−13} Herein, we develop [a ne](#page-6-0)w method to achieve oil−water separation, which depends on the hydration effect indu[ced b](#page-6-0)y a large electric dipole. When the composite nanoparticles were exposed to the water droplets in an oil−water mixture, they swelled strongly and absorbed the water droplets like a sponge.

Water molecules can form hydration layers surrounding charges in aqueous media, which is attributed to the large electric dipole by virtue of the residual charges on the H and O atoms.14,15 Such a hydration layer greatly lowers the self-energy of the entrapped charge, that is, it can be very difficult to perma[nent](#page-6-0)ly remove a water molecule from the hydration layer surrounding a charge.^{16,17} In this case, water molecules were still firmly absorbed around a charge in spite of vigorous shock during the practical [oil](#page-6-0)[−](#page-6-0)water separation process. Magnetic polymeric nanoparticles, a promising advanced composite material, have attracted tremendous attention owing to their unique magnetic responsiveness and their potential applications in various fields, such as bioseparation,18−²¹ magnetically targeted drug delivery, $22,23$ and magnetic resonance imaging.24−²⁶ In particular, designing functional [magne](#page-6-0)tic polymeric nanoparticles with speci[al we](#page-6-0)ttability is of great concern in the fiel[d](#page-6-0) [of](#page-6-0) oil/water separation.²⁷ Therefore, charged polymer brushes-grafted magnetic nanoparticles were prepared herein to combine the advantages of c[ha](#page-6-0)rged polymer brushes on the hydration effect and magnetic nanoparticles on facile separation and large surface area, making our present core−shell−corona nanoparticles promising oil-water separation materials.²⁸⁻³⁰ As

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Scheme 1. Illustration of the Synthetic Procedure for the Charged Polymer Brushes-Grafted Magnetic Core−Shell−Corona Composite Nanoparticles (Fe₃O₄@SiO₂@PSPMA), Hydration Mechanism of PSPMA Brushes, and Procedure for Water Harvesting and Oil/Water Separation

proof-of-concept, negatively charged poly(3-sulfopropyl methacrylate potassium salt) (PSPMA) brushes-grafted $Fe₃O₄(@SiO₂)$ composite nanoparticles were prepared. PSPMA brushes possess strong hydration capability in aqueous media^{31,32} that can harvest water droplets in an oil−water mixture. Use of a silica coating is to protect $Fe₃O₄$ cores from da[mage](#page-7-0) and aggregation and supplies a suitable supporting matrix to immobilize functional groups owing to numerous surface Si− OH groups.33−³⁶ When the composite magnetic nanoparticles were put into an oil−water mixture, the dispersed water was gathered to[ge](#page-7-0)t[he](#page-7-0)r to form hydrated liquid marbles, and then the marbles could freely move using an external magnetic field to realize separation.

2. EXPERIMENTAL SECTION

2.1. Materials. Ferric chloride hexahydrate (FeCl₃·6H₂O), ethylene glycol (EG), anhydrous sodium acetate, and ammonium hydroxide (25−28%) and methyl blue were purchased from Sinopharm Chemical Reagent Co., Ltd. (China). Tetraethyl orthosilicate (TEOS, 98%) was purchased from Sigma-Aldrich. 3-(Trichlorosilyl)propyl-2-bromo-2 methylpropanoate (initiator ATRP-Br) was synthesized in our lab.37[−]³⁹ 3-Sulfopropyl methacrylate potassium salt (98%, SPMA), 2,2′-bipyridine (Bpy, 99%), and copper(I) bromide (CuBr) were pu[rchase](#page-7-0)d from TCI Co., Ltd. CuBr was refluxed with acetic acid prior to use. Toluene was purified by distillation over CaH₂. Other chemicals were used as received. Deionized water was used for all polymerization and treatment processes.

2.2. Synthesis of Fe₃O₄ Magnetic Nanoparticles. Fe₃O₄ magnetic nanoparticles were prepared through a modified solvothermal reaction.⁴⁰ Briefly, 1.35 g of FeCl₃·6H₂O, 3.6 g of sodium acetate, and 0.4 g of sodium citrate were dissolved in 50 mL of ethylene glycol via stirring. [Th](#page-7-0)e obtained homogeneous yellow solution was then sealed in a Teflon-lined stainless-steel autoclave (100 mL) and heated

at 200 °C for 12 h. After being cooled to room temperature, the obtained black magnetite particles were collected, washed with ethanol several times, and finally dried in vacuum at 40 °C for 24 h.

2.3. Synthesis of Fe₃O₄@SiO₂ Core–Shell Nanoparticles. Fe₃O₄@SiO₂ core−shell nanoparticles were prepared according to the method previously reported with slight modification.⁴¹ Typically, 50 mg of Fe3O4 nanoparticles was dispersed in a mixture of 40 mL of ethanol and 10 mL of water by ultrasonication for 20 [min](#page-7-0). Then under continuous mechanical stirring, 1.5 mL of ammonia solution (25−28 wt %) was added to the mixture, followed by 0.3 mL of TEOS being added dropwise. After being kept at room temperature for 6 h, the solid products were collected using an external magnet, rinsed with water and ethanol, and dried in a vacuum oven at 40 °C for 12 h.

2.4. Immobilization of ATRP Initiator. Fe₃O₄@SiO₂ nanoparticles (25 mg) were stirred at room temperature for 24 h in the anhydrous toluene solution (25 mL) containing 10 μ L of 3-(trichlorosilyl)propyl 2-bromo-2-methylpropanoate. The hydrolysis/ condensation process between the ATRP initiator and silica resulted in the immobilization of initiator. Then the initiator-immobilized $Fe₃O₄(\partial SiO₂)$ nanoparticles were collected by a magnet, washed several times with fresh anhydrous toluene and ethanol, and dried in vacuum at room temperature for 24 h.

2.5. Synthesis of PSPMA Brushes-Grafted Fe₃O₄@SiO₂ Composite Nanoparticles by Surface-Initiated ATRP. Typically, the initiator-immobilized $Fe₃O₄(@SiO₂$ nanoparticles (10 mg) and SPMA (1.5 g) were dispersed in 1:4 (v:v) MeOH/H₂O (10 mL) via 10 min of ultrasonication. After being deoxygenated for 30 min at room temperature using nitrogen flow, 2,2′-bipyridyl (0.0625 g, 0.4 mmol) and CuBr (0.0287 g, 0.2 mmol) were added into the dispersion quickly to trigger polymerization. Polymerization was conducted at room temperature under N_2 protection for 2 h. Finally, PSPMA brushes-grafted $Fe₃O₄(@SiO₂$ composite nanoparticles were collected by a magnet, washed several times with deionized water and ethanol, and dried in a vacuum oven at room temperature for 24 h.

Figure 1. TEM image (a) and SAED pattern (b) of Fe₃O₄ nanoparticles; TEM images of Fe₃O₄@SiO₂ core−shell particles (c) and Fe₃O₄@SiO₂@ PSPMA composites (d, e); EDXS spectra of $Fe₃O₄(@SiO₂(@PSPMA composites (f)).$

2.6. Water Harvesting and Oil/Water Separation. Scheme 1 illustrates the procedure for water harvesting and oil/water separation. Water droplets were dyed blue by methyl blue. For the water droplet floating on the oil surface, the composite particles were placed on t[he](#page-1-0) water droplet. After a water-absorbing process (about 30 s), the particles/water marble formed and was driven by a regular magnet. For the water droplets under oil, the magnetic particles were added into the oil/water mixture and shaken vigorously. After formation of particles/water marble, the marble was moved to the magnet responsively.

2.7. Characterization. Transmission electron microscopy (TEM) images were obtained from a FEI Tecnai G2 TF20 transmission electron microscope with a field emission gun operating at 200 kV. The elemental composition was evaluated using energy-dispersive Xray spectroscopy (EDXS). Fourier transform infrared (FT-IR) spectra were determined on a Nicolet iS10 (Thermo Scientific, USA) FT-IR spectrometer. Thermogravimetric analysis (TGA) was performed on a STA 449C TG-DSC instrument. X-ray diffraction (XRD) measurements were carried out on a PANalytical X'Pert PRO X-ray diffractometer with Cu Ka (λ = 0.15418 nm) incident radiation. Magnetic properties were studied on a vibrating sample magnetometer (VSM) on a model 6000 physical property measurement system (Quantum, USA) at 300 K. Hydrodynamic diameters (D_h) and zeta potential were measured on a particle size analyzer (Zetasizer Nano ZS, Malvern Instruments, UK) equipped with a 632.8 nm He−Ne laser by dynamic light scattering (DLS) technique. The swelling ratio (SR) was deremined by the following equation: $SR = V_{\text{aqueous}}/V_{\text{dry}} =$ $(D_{h}/D_{\text{dry}})^3$ where V_{aqueous} and V_{dry} were the volumes in aqueous media and in the dry state, respectively, and D_{aqueous} and D_{dry} were the mean diameters [fr](#page-6-0)om DLS and TEM, respectively. The resulting composite particles were deposited on a microfiltration membrane (pore size 220 nm) by filtration and dried and form a dense and thick film. The water contact angle on the deposited film was measured with a 5 μ L droplet of distilled water at ambient temperature with a DSA100 contact angle (CA) meter (Kruss Co., Germany). The oil contact angle on the

deposited film was tested using a 5 μ L droplet of dichloromethane dyed by oil red in water. In the reusability test, 10 mg of the nanocomposites and 30 mg of water were adopted. The used nanocomposites were centrifuged and dried before the next cycle. KCl solutions with different concentrations were employed to investigate the effect of ionic strength on the nanocomposites. PBS buffer solutions with a pH value from 2 to 14 were prepared to investigate the effect of pH on the nanocomposites.

3. RESULTS AND DISCUSSION

3.1. Synthesis and Characterization of $Fe₃O₄@SiO₂@$ PSPMA Nanoparticles. The procedure for preparation of PSPMA brushes-grafted Fe₃O₄@SiO₂ core−shell−corona composite nanoparticles (Fe₃O₄@SiO₂@PSPMA) is illustrated in Scheme 1. Briefly, magnetic $Fe₃O₄$ NPs were first prepared via a solvothermal reaction. Then, the $Fe₃O₄$ NPs were encapsulate[d](#page-1-0) by a $SiO₂$ shell through a sol−gel process. Finally, core−shell−corona structured nanoparticles, PSPMA brushesgrafted $Fe₃O₄(@SiO₂)$, were obtained via immobilizing Brcontaining initiator, followed by surface-initiated ATRP of SPMA.

Representative morphology of the as-prepared particles $(Fe₃O₄, Fe₃O₄(@SiO₂, and Fe₃O₄(@SiO₂(@PSPMA)) was ob$ served by TEM, which verifies the synthetic procedure illustrated in Scheme 1, as shown in Figure 1. It showed that the $Fe₃O₄$ particles prepared via solvothermal reaction were nearly spherical in sh[ap](#page-1-0)e with mean diameter of ca. 356 nm (Figure 1a). The magnetite exhibits polycrystalline features that were confirmed by spotty diffraction rings of selected area electron diffraction pattern (Figure 1b). Then, as observed by the TEM image in Figure 1c, a layer of dense amorphous silica with a thickness of ca. 55 nm was successfully coated on the

Fe3O4 particles via the sol−gel process, resulting in the magnetic core−shell Fe₃O₄@SiO₂ microspheres. Afterward, the surface-initiated ATRP resulted in a core−shell−corona structure of $Fe₃O₄(@SiO₂(@PSPMA (Figure 1d). The magnified$ image shown in Figure 1e evidently reveals a dark $Fe₃O₄$ core, a gray $SiO₂$ middle shell, and a light-gray PS[PM](#page-2-0)A corona about 42 nm in thickness. S[uc](#page-2-0)cessful grafting of PSPMA brushes on the surface of $Fe₃O₄(ω SiO₂ was also confirmed by the EDXS$ measurement (Figure 1f), which shows the appearance of S and K signals attributed to the PSPMA.

Successful prepara[tio](#page-2-0)n of $Fe₃O₄(@SiO₂(@PSPMA was also$ confirmed by FTIR spectroscopy (Figure 2). For $Fe₃O₄$ NPs

Figure 2. FTIR spectra of (a) Fe₃O₄, (b) Fe₃O₄@SiO₂, (c) Fe₃O₄@ SiO₂−Br, and (d) Fe₃O₄@SiO₂@PSPMA.

(Figure 2a), the absorption peak at 587 cm⁻¹ was assigned to Fe−O absorption. For Fe₃O₄@SiO₂ (Figure 2b), the absorption peak of Si−O−Si appeared at 1091 cm[−]¹ whereas the wide peak at 3400 cm^{-1} was assigned to the stretching absorption vibration of Si−OH. After modification with ATRP initiator, some new peaks were observed from $Fe₃O₄(@SiO₂−Br)$ (Figure 2c). The peaks at 1720 and 2940 cm^{-1} can be assigned to the ester C=O and C−H bonds from the silane initiator. After the PSPMA brushes were grafted from the surface of Fe3O4@SiO2−Br via SI-ATRP, the characteristic absorption peaks of S= O in SO_3^- groups appeared at 1190 and 1047 cm[−]¹ , whereas the strong absorption peak at 1720 cm[−]¹ appeared from $C=O$ in ester groups (Figure 2d). The appearance of these absorption peaks confirmed that polymer brushes have been grafted onto Fe₃O₄@SiO₂−Br.

To quantitatively determine the composite of the obtained $Fe₃O₄(@SiO₂(@PSPMA, thermogravimetric analysis was con$ ducted. Figure 3 showed the TGA curves of the obtained Fe₃O₄, Fe₃O₄@SiO₂, Fe₃O₄@SiO₂–Br, and Fe₃O₄@SiO₂@ PSPMA, and the weight loss was evaluated from 50 to 800 °C. The weight loss of $Fe₃O₄$ was 3.8 wt %, which was attributed to degradation of the stabilizer sodium citrate (Figure 3a). A weight loss of 6.9 wt % was observed from the curve of Fe_3O_4 @ $SiO₂$ (Figure 3b), which was related to water molecules bonded to $SiO₂$. After immobilization of ATRP initiator, the weight loss of Fe₃O₄@SiO₂−Br was 9.7 wt % (Figure 3c) and the content of the initiator was calculated to be 2.8 wt %. Moreover, after being grafted with PSPMA brushes by ST-ATRP, the weight loss of composite particles was increased to 33.8 wt %, indicating the content of PSPMA brushes was 24.1 wt % (Figure 3d). Notably, from the TGA curve of $Fe_3O_4@SiO_2@$ PSPMA there were two obvious decomposition steps. The first

Figure 3. TGA curves (a) of Fe₃O₄, (b) Fe₃O₄@SiO₂, (c) Fe₃O₄@ SiO₂−Br, and (d) Fe₃O₄@SiO₂@PSPMA.

step at around 300 °C was derived from conventional thermal decomposition of a majority of PSPMA brushes. The second step at around 680 °C was attributed to the weight loss of partial PSPMA brushes that were directly attached to the surface of $Fe₃O₄(@SiO₂)$ nanoparticles. The second decomposition step reflects the strong attachment between $Fe₃O₄(a)$ $SiO₂$ and PSPMA brushes, which is similar to previous reports.19,42,43

Crystal structures of the as-prepared $Fe₃O₄$, $Fe₃O₄(@SiO₂)$ and Fe₃O₄@SiO₂@PSPMA were also investigated by X-ray diffraction (XRD) and are shown in Figure 4.The diffraction

Figure 4. XRD patterns for (a) Fe₃O₄, (b) Fe₃O₄@SiO₂, and (c) $Fe₃O₄(@SiO₂(@PSPMA.$

peaks can be indexed to the face-centered cubic structure of magnetite according to JCPDS.⁴⁴ Six characteristic peaks (2θ = 30.3°, 35.6°, 43.2°, 53.5°, 57.2°, and 62.9°) related to their corresponding indices ((220), [\(31](#page-7-0)1), (400), (422), (511), and (440)) were observed in the case of the $Fe₃O₄$ magnetic particles (Figure 4a). Then, for an XRD pattern of the $Fe₃O₄(\omega)$ SiO₂ (Figure 4b), a broad peak at 2 θ of 22−26° corresponded to the amorphous peak of $SiO₂$. Furthermore, after the coverage of PSPMA brushes, another broad diffusion pattern of amorphous polymer layer appeared at 2θ of 16−20° in the XRD pattern of $Fe₃O₄(@SiO₂(@PSPMA (Figure 4c)).$

3.2. Magnetic Property of the As-Prepared Nanoparticles. The magnetic property of the as-prepared nanoparticles was investigated by a vibrating sample magnetometer at room temperature. Figure 5 showed the magnetic hysteresis

Figure 5. Magnetic hysteresis loops of (a) Fe₃O₄, (b) Fe₃O₄@SiO₂, and (c) $Fe₃O₄(@SiO₂(@PSPMA. (Inset) Picture of $Fe₃O₄(@SiO₂(@PSPMA. (Inset))$$ PSPMA gathered and driven by a regular magnet in water.

loops of Fe₃O₄, Fe₃O₄@SiO₂, and Fe₃O₄@SiO₂@PSPMA, which exhibited superparamagnetic behavior and no obvious remanence and coercivity.⁴⁵ The magnetic saturation (M_s) values were 69.5, 31.3, and 15.2 emu/g, respectively. The M_s values of $Fe₃O₄(@SiO₂$ an[d](#page-7-0) $Fe₃O₄(@SiO₂(@PSPMA are both$ lower than that of $Fe₃O₄$, which is due to the decrease in the density of $Fe₃O₄$ in the obtained nanocomposites after coverage of nonmagnetic $SiO₂$ and grafting of nonmagnetic PSPMA brushes. It should be noted that the resulting composite nanoparticles still showed strong magnetization, which suggested their suitability for magnetic separation and targeting using a regular magnetic plate (Figure 5, inset).

3.3. Hydration Capability of $Fe₃O₄@SiO₂@PSPMA$ Nanocomposites. To investigate the hydration capability of PSPMA brushes, the hydrodynamic diameter and zeta potential of Fe₃O₄@SiO₂@PSPMA particles was measured using DLS in aqueous media. As shown in Figure 6a, the intensity-average hydrodynamic diameter of Fe₃O₄@SiO₂@PSPMA (ca. 864 nm) was much larger than the size obtained in the TEM images (ca. 550 nm). Correspondingly, the swelling ratio $(V_{\text{aqueous}}/$ V_{dry}) was calculated to be 3.88. Because the ionic PSPMA polymer chains could be fully extended in aqueous media, $Fe₃O₄(\omega$ SiO₂ $(\omega$ PSPMA composites possessed a good swelling capability like a sponge. On the basis of the hydration mechanism illustrated in Scheme 1, water molecules could form hydrated layer surrounding a charge. To lower the born energy of the encapsulated polar gro[up](#page-1-0)s, the hydrated layer was robust and difficult to be removed. As shown in Figure 6b, the zeta potential of Fe₃O₄@SiO₂@PSPMA particles is −53.2 mV, suggesting a strong hydration capability of negatively charged PSPMA brushes. Additionally, because of electrical interaction, the hydrodynamic size and zeta potential distribution both exhibited a single peak, indicating a good dispersibility of the resultant composites in aqueous media.

Due to the strong hydration capacity and good swelling performance of PSPMA brushes, the $Fe₃O₄(@SiO₂(@PSPMA$ composites exhibited special wettability. As shown in Figure 6c and Video S1 (Supporting Information), the contact angle (CA) of the water dropped on the $Fe₃O₄(@SiO₂(@PSPMA film$ was reduced to 0° [within ca. 25 s, suggestin](#page-6-0)g a superhydrophilic performance of $Fe₃O₄(@SiO₂(@PSPMA)$ nanoparticles. Interestingly, the water droplet permeated into the interior of the film quickly rather than spreading on the film, which demonstrates that the as-prepared composite nanoparticles possess strong water-absorbing capability. This phenomenon was attributed to the strong hydration effect and good swelling performance of the outermost PSPMA brushes. In addition to the strong waterabsorbing capability, the wettability of the Fe₃O₄@SiO₂@ PSPMA film for oil under water was also investigated. As shown in Figure S1 (Supporting Information), the contact angle of dichloromethane on the film underwater was $165 \pm 3^{\circ}$, indicating a s[uperoleophobic propert](#page-6-0)y. Thus, in oil/water mixture, the nanocomposites can be fast wetted by water instead of oil. Moreover, since the PSPMA brushes-grafted $Fe₃O₄(@SiO₂)$ nanoparticles possess a strong magnetic core and a superhydrophilic PSPMA exterior, they hold great promise for rapid magnetic separation of water in oil/water systems.

3.4. Water Harvesting and Oil−Water Separation. On the basis of its selective wettability, the obtained PSPMA brushes-grafted magnetic particles were used to absorb water from oil−water mixture, and the schematic diagram is illustrated in Scheme 1. In general, the distribution of water droplets in the oil has two modes: one is the water droplets floating on the oil surf[ac](#page-1-0)e; the other is the water droplets under

Figure 6. (a) Hydrodynamic diameter distribution of Fe₃O₄@SiO₂@PSPMA particles. (b) Zeta-potential distribution of Fe₃O₄@SiO₂@PSPMA particles. (c) Changing contact angel of water on the deposited film of $Fe₃O₄(@SiO₂(@PSPMA particles with time.$

Figure 7. Water harvesting and oil/water separation procedures for the water droplets floating on dichloromethane surface (a) and the water droplets under hexane (b).

oil. Hopefully, the two types of water droplets are both harvested by the as-prepared composite magnetic particles.

Dichloromethane/ H_2O mixture was chosen to investigate the ability of the as-prepared composite nanoparticle to harvest the water droplets floating on the oil surface. As shown in Figure 7a, a drop of water, which has been dyed blue, was dropped on the oil surface, and then the superhydrophilic magnetic particles were added and spread in the water. Due to the high hydration capability of PSPMA brushes, the water was absorbed and surrounding the PSPMA brushes-grafted magnetic particles. Subsequently, the particle/water marble was formed. When putting a magnet close by, the water-holding magnetic particles moved along with the magnet responsively. Fortunately, these particles can be recycled by centrifugation. Additionally, the isolated water-holding magnetic nanoparticles could be gathered together and removed through the magnet (Video S2, Supporting Information), which broadened the range of water collection. Using a certain amount of nanocomposites (10 [mg\), up to 50 mg of wa](#page-6-0)ter can be absorbed and removed by a magnet. Besides, the reusability of the nanocomposites was studied by a cycle test. The used nanocomposites was first centrifuged and dried. As shown in Table S1 (Supporting Information), in spite of a small loss, the water harvesting process of the dried nanocomposites can be success[fully cycled.](#page-6-0)

[Furthermo](#page-6-0)re, hexane/water mixture was chosen to test the absorption capability of the as-prepared composite nanoparticles to harvest the water droplets under oil. As shown in Figure 7b, a drop of water, which has been dyed by methyl blue, was dropped into the hexane. Then the magnetic particles were added into the oil/water mixture and shaken vigorously. Afterward, the small dispersed water droplets were absorbed around the PSPMA brushes-grafted magnetic particles. Interestingly, the small dispersed PSPMA brushes-grafted magnetic particles with absorbed water can gather together and form a big magnetic bead through a slight shaking, which may be attributed to the high surface energy of $Fe₃O₄(@SiO₂(@)$ PSPMA nanoparticles, easily triggering aggregation of the nanoparticles. Due to its magnetic property, the PSPMA brushes-grafted magnetic particles with absorbed water could also be facilely moved under external magnetic control. When putting a magnet close by, the magnetic bead moved toward the magnet responsively. In addition, the isolated water-holding magnetic beads, which were formed by vigorous shaking, could gather together fast through liquid−liquid separation (Video S3, Supporting Information). Notably, the "water marble" was

spherical and could roll freely under the consideration of minimizing the surface free energy (Video S4, Supporting Information). These results demonstrated that the water droplets could be easily harvested by $Fe₃O₄(@SiO₂(@PSPMA$ [and separate](#page-6-0)d using a regular magnet.

To investigate the effect of ionic strength and pH on the water-absorbing materials for harvesting water, the hydrodynamic size of $Fe₃O₄(@SiO₂(@PSPMA nanocomposites under$ various ionic strength and pH was first measured by DLS. As shown in Figure S2 (Supporting Information), the hydrodynamic diameter decreased gradually with increasing concentration of KCl solution[, whereas the hydrodyna](#page-6-0)mic diameter had no obvious change with pH value. The conformation transition of polyelectrolyte brushes can be adjusted by different pH and ion strength.⁴⁶ In pure water, the PSPMA brushes show a fully stretched conformation because of electrostatic repulsion. In [KC](#page-7-0)l electrolyte solution, the polyelectrolyte brushes tend to collapse because of strong electrostatic screening, and part of water was excluded from PSPMA brushes shown in the inset of Figure $S_2(a)$ (Supporting Information.⁴⁷ Besides, because SPMA was assigned to strong electrolyte, the change of pH has no obvious eff[ect on the PS](#page-6-0)[PM](#page-7-0)A brushes. This factor has a remarkable effect on the water-absorbing materials for harvesting water. With increasing concentration of KCl solution, more and more $Fe₃O₄(@SiO₂(@PSPMA nanocomposites)$ were needed to harvest a certain amount of KCl solution (50 mg). In contrast, the required amount of nanoparticles had no obvious change as the pH value increased. The effect of ionic strength and pH on the water-absorbing materials had a guiding significance for practical application.

By introduction of charged PSPMA polymer brushes with strong hydration capability, the Fe₃O₄@SiO₂@PSPMA nanocomposites can achieve a good water-absorbing performance like a sponge. Taking advantage of magnetic $Fe₃O₄$ cores, $Fe₃O₄(@SiO₂(@PSPMA composites with absorbed water can be$ separated intelligently from the oil/water mixture by an external magnetic field.

4. CONCLUSION

In summary, we developed a novel material, charged polymer brushes-grafted magnetic core−shell−corona composite nanoparticles (Fe₃O₄@SiO₂@PSPMA), to harvest water through the hydration effect and achieve oil−water separation driven by an external magnetic field. The charged polymer brushes, PSPMA brushes, have both strong hydration capability and good swelling performance, which endow the as-prepared composite nanoparticles with superhydrophilicity and a good waterabsorbing performance like a sponge. The $SiO₂$ shell not only protects $Fe₃O₄$ cores from damage but also supplies a suitable supporting matrix to immobilize functional groups through numerous surface Si−OH groups. Importantly, the Fe₃O₄@ $SiO₂(\omega)$ PSPMA nanoparticles with absorbed water can be separated from the oil driven by an external magnetic field. This water-absorbing material uses selective wettability to harvest water and achieve oil−water separation, which is believed to be useful in finding novel approaches for recycling water from sewage and removing water in the petroleum industry.

■ ASSOCIATED CONTENT

6 Supporting Information

Wettability of the deposited film, water harvesting in dichloromethane, formation of water-holding magnetic beads, water harvesting in hexane, contact angle of an oil droplet on the deposited film in water, cycle test for studying the reusability of the nanocomposites, variation of hydrodynamic diameter, and required amount of nanoparticle as a function of ionic strength or pH, and videos demonstrating the performances of the resultant nanoparticles. This material is available free of charge via the Internet at http://pubs.acs.org.

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